

1 **PORE MORPHOLOGY VARIATION UNDER AMBIENT CURING OF PLAIN AND FIBER-REINFORCED**
2 **HIGH PERFORMANCE MORTAR AT AN EARLY AGE**

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11 Keywords: pore morphology, plain mortar, fiber reinforced mortar, CT-Scan technology.

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14 **ABSTRACT**

15 Pore morphology and its changes at an early age in two different mortar mixtures are analyzed by means of a CT-
16 Scan and DIP software. Both mixtures share the same cement paste composition. The only difference between them is the
17 inclusion of 0.1% by volume of steel fibers in one paste and its omission from the other paste. A total of six specimen
18 cylinders measuring 45.2 mm in diameter and 50 mm in height were cast. All the specimens were held in a controlled
19 atmosphere at 20°C and at 60% humidity, reflecting natural ambient curing conditions. Each specimen was scanned at 1, 2,
20 3, 4, and 7 days using a micro CT-Scan. The data were analyzed using a digital image processin (DIP) software and some
21 post-processing subroutines. Each individual void was identified and isolated and all its geometrical parameters were
22 measured. Among the most interesting and relevant findings was that the presence of fibers substantially modified the pore
23 morphology, increasing the volume of voids, and the pore-size, and reducing the shape factor of the voids, among other
24 effects. Both mixtures showed different pore morphologies at the beginning of the curing process that metamorphosed into
25 different and even divergent geometrical dimensions.

26

27 **1. INTRODUCTION**

28 All concrete elements contain a certain percentage of voids, which differ greatly from one piece to another. In
29 general, concrete porosity is residual and non-desirable. However, in other cases, concrete design entails the use of air-
30 entrainment agents to have a specific percentage porosity.

31 The voids content has a strong influence on the macroscopic behavior of concrete at all ages. In case of fresh
32 concrete, the content of voids modifies its rheology [1-3]. Thus, for example, an increase in the voids content results in a
33 reduction of concrete viscosity, which is very important in case of pumpable concretes [4]. In case of hardened concrete, the
34 voids content influences many aspects, such us permeability [5-7]. Another interesting case is pervious concrete, designed
35 with a very high porosity, so that, among other applications, road surfaces can, even in heavy rain, remain relatively dry [5,
36 7].

37 The strong relation may also be highlighted between voids content and behavior under freeze-thaw cycles [8, 9]. In
38 fact, concrete subjected to very intense freeze-thaw cycles must be designed with a minimum threshold of voids content.
39 Additionally, the voids content also influences concrete behavior at high temperatures [10] and its behavior under cyclic
40 loading, among other aspects [11-13].

41 Depending on the void size, they can be classified into micropores (smaller than 1 μm), mesopores (between 1 μm
42 and 10 mm) and macropores, as well as macropores or simply pores (larger than 10 mm) [14]. Several methods are used to
43 analyze voids. The traditional ones are nitrogen absorption and mercury-intrusion porosimetry (MIP) [15, 16]. These

44 traditional methods show two main limitations. First, they can only provide pore-size distribution, but not the pore
45 distribution, shape, etc. Second, these techniques can only provide information on open porosity, but not on closed porosity.

46 A novel technology has recently been successfully applied to concrete: computerized tomography (CT) scan
47 technology. Beyond the use of CT scans in medicine, a lot of research has recently been conducted on the analysis of
48 internal concrete microstructure. Most have focused on fiber-reinforced concrete and therefore on fiber orientation [12, 17,
49 18]. However, interest has, over the past few years, been focused on internal voids with several works published on the
50 topic [4, 5, 19-26].

51 In case of fiber-reinforced concrete, fibers have a strong influence on the macroscopic behavior of concrete at all
52 ages. Fibers are commonly used to improve the mechanical behavior of hardened concrete. For example, they are
53 commonly used to reduce the cracking, which improves the durability of the structure [27, 28]. They are also used to
54 improve fatigue life [29-31], to increase tensile strength [32, 33], and to improve behavior under freeze-thaw cycles [34,
55 35], among other uses.

56 The inclusion of fibers in fresh concrete modifies its behavior, by increasing concrete paste viscosity [36-40].

57 However, all the research mentioned above implicitly assume that fibers do not modify the concrete matrix. This
58 work demonstrates that steel fibers significantly modify the voids content of concrete and its porosity. A fiber presence
59 results in a different evolution of internal porosity during the early age of concrete and likewise in hardened concretes that
60 reveals different patterns of internal voids. In recent years, some interesting research projects have been carried out to study
61 the interfacial zone between the fibers and the cement paste, in order to analyze whether, at this level, a variation in the
62 cement paste composition occurs. In all cases, the nano-indentation technique is used; usually combined with scanning
63 electron microscopy (SEM) [41-43]. The results show differences in the composition of the cement paste in this zone,
64 although its influence on the macroscopic behavior of the material has not been deeply studied.

65 Mortar is a particular case of concrete, where no coarse aggregate is used. All the issues described above are also
66 applicable to mortar.

67 In this work, two different mortar mixtures have been studied: plain and steel fiber-reinforced mortar. The only
68 difference between each is the inclusion or otherwise of fibers. All the specimens have been scanned using a CT-Scan to
69 detect the voids. Using post-processing routines, especially developed by the authors, the pore morphology in both mixtures
70 has been observed and compared: porosity, pore-size distribution, pore shape, etc. Additionally, the variation during the
71 early age has been studied and a very different change has been detected.

72 The objective of this work is to analyze the change of the pore morphology and the main geometrical parameters of
73 the voids of these two mortar mixtures over time during the first curing week, and how the presence of fibers modifies this
74 change over time.

75 The structure of the paper is as follows: the experimental procedure is presented in Section 2; the results of the tests
76 are described and discussed in Section 3; and finally, the conclusions are shared in Section 4.

77

78 2. EXPERIMENTAL PROGRAM

79 The materials, the manufacturing procedure and the scanning procedure are described in this section.

80 2.1 Materials

81 In this study, a total of six cylinders were cast. The specimen dimensions were 45.2 mm in diameter and 50 mm in
82 height. Three of them were made of steel fiber-reinforced mortar (SFRM) and three of plain mortar (PM). Table 1 shows the
83 mixture proportions.

84

Table 1: Mixture proportions

Materials	PM	SFRM
Cement (kg/m ³)	700.0	
Water (kg/m ³)	217.0	
Superplasticizer (kg/m ³)	24.5	
Nanosilica (kg/m ³)	10.5	
Fine aggregate (0/4) (kg/m ³)	1,400.0	
Fiber (% by volume)	0.0%	0.1%

85

86 An amount of 7.8 kg/m³ of steel Dramix 8/.16 (BEKAERT, Kortrijk, Belgium), fiber were added to manufacture the
87 fiber-reinforced mortar, resulting in a fiber volume fraction of 0.1%. The fibers measured 8 mm in length and 0.16 mm in
88 diameter and had an aspect ratio of 29. According to the supplier information, the tensile strength was 3000 MPa and the
89 modulus of elasticity was 200 GPa. The mortar mass included MasterRoc MS 685 (BASF, Ludwigshafen am Rhein,
90 Germany) nanosilica, Glenium 52 (BASF, Ludwigshafen am Rhein, Germany) superplasticizer and siliceous aggregate,
91 with a nominal maximum aggregate size of 4 mm that were all mixed with a high strength Portland cement CEM I 52.5 R.

92 It can be noticed in Table 1 that mortar has been considered in this work, instead of concrete. This is because the CT-
93 Scan demands small-sized specimens to obtain accuracy enough to detect properly the voids. In this situation, the use of
94 coarse aggregate in small specimens results in a high scatter between the different specimens, which complicate the further

95 analysis. In addition, the dimensions of the specimen are chosen to assure a successful scanning process, with high
96 accuracy.

97 Additionally, 2 prisms 40x40x160 mm were performed; one per mixture. A total of 3 cubes with 40 mm edges were
98 obtained from each prism and tested under compression in order to characterize their compression strength (according to
99 [44]). The average compressive strength f_c was 68.6 MPa for PM and 69.2 MPa for SFRM, with a standard deviation of 1.7
100 MPa and 1.9 MPa respectively. From a statistical point of view, no difference between PM and SFRM can be observed
101 regarding compressive strength. The mean mortar density ρ was 2343 kg/m³ and 2348 kg/m³ for PM and SFRM
102 respectively, with a standard deviation of 23.3 kg/m³ and 23.5 kg/m³ respectively. Again, from a statistical point of view,
103 there is no difference between both mortars regarding density.

104 The viscosity of the fresh mortar was measured, for both mixtures, by flow table according to [45]. The results
105 obtained were 240 mm for PM and 210 for SFRM. In both cases the mixtures can be considered as fluid, although PM show
106 more flowability than SFRM.

107 **2.2 Specimen manufacturing process**

108 PVC molds with an inner diameter of 45.2 mm, an outer diameter of 50 mm and a height of 50 mm were used to cast
109 the specimens. The PVC molds were built using a commercial PVC pipe. The base of each cast was a PVC disc with a
110 diameter of 60 mm and a thickness of 3 mm. The base was welded to the pipes, in order to assure a watertight joint (Figure
111 1). The mortar was prepared in a cement mortar mixer, following the manufacturing specifications in standard EN 196-
112 1:2016. The protocol of mixing was the same for the two mixtures. First, the dry components were poured on the mixer and
113 finally the water and the superplasticizer were added. In case of the SFRM, finally, the fibers were added. The molds were
114 filled in two parts using a small aluminum scoop to form the specimens without applying vibration. However, some small
115 punches were applied on the side of the molds, once it was filled with mortar, to help mortar to expel the entrapped air.
116 Finally, the upper surface was smoothed with a trowel. Once cast, all the specimens were held under controlled conditions
117 at 20°C and 60% humidity, reflecting ambient curing conditions more closely than a conventional curing room, where
118 specimens remain at 20°C and 100% humidity.

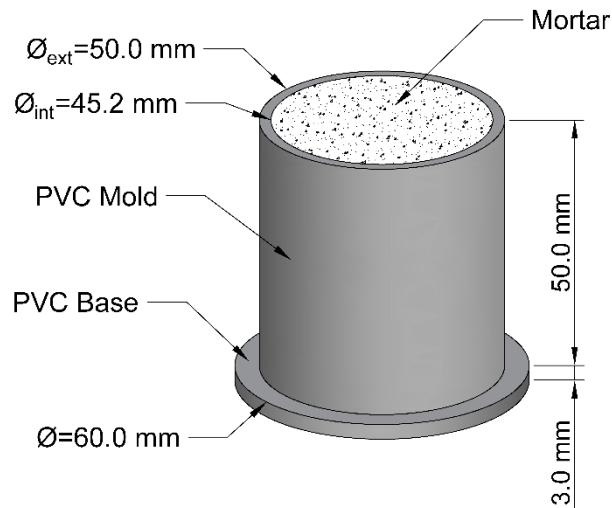


Figure 1: Specimen and mold

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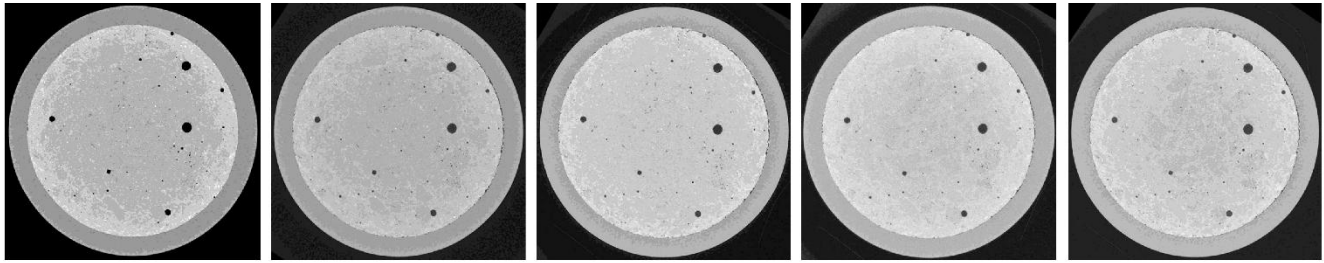
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121 2.3 Scanning process

122 This study is focused on the behavior of mortar during an early age, in particular during the first week of curing
 123 where the most relevant changes in the microstructure of mortar are expected to occur. To do so, all the specimens were
 124 scanned daily on the 1st, 2nd, 3rd, 4th and 7th day, using a micro-CT Scan.

125 A GE Phoenix v|tome|x CT system equipped with a 300 kV/500W nano-focus x-ray tube was used at the ‘*Centro*
 126 *Nacional de Investigación sobre la Evolución Humana (CENIEH)*’, (Burgos, Spain). The accelerating voltage was 160 kV
 127 and the current was 150 μ A. The CT-Scan has post-processing software that generates 2D images with 2048x2048 pixels.
 128 Thus, the equipment provides a horizontal resolution of $25 \times 25 \mu\text{m}^2$ for a section with a diameter of 45.2 mm. The vertical
 129 distance between the cutting planes was fixed at 25 μm , so the CT-Scan produced 2,000 images per specimen, such as those
 130 shown in Figures 2 and 3. The voxel size was $25 \times 25 \times 25 \mu\text{m}^3$. The post-processing software created a 3D image of the
 131 specimens using all of the above-mentioned images. The software assigns a grey level to each voxel (volumetric pixel),
 132 varying from 0 to 255 (where 0 is equal to black and 255 is equal to white), depending on the real density of the matter at
 133 that point. Light grey voxels correspond to denser points and dark grey ones, to less dense ones, i.e., voids. The output of all
 134 this process was a matrix that included the X, Y, and Z coordinates of the voxel center of gravity and a number, from 0 to
 135 255, referring to the density. The total number of voxels in each specimen were approximately $4.3 \cdot 10^9$.

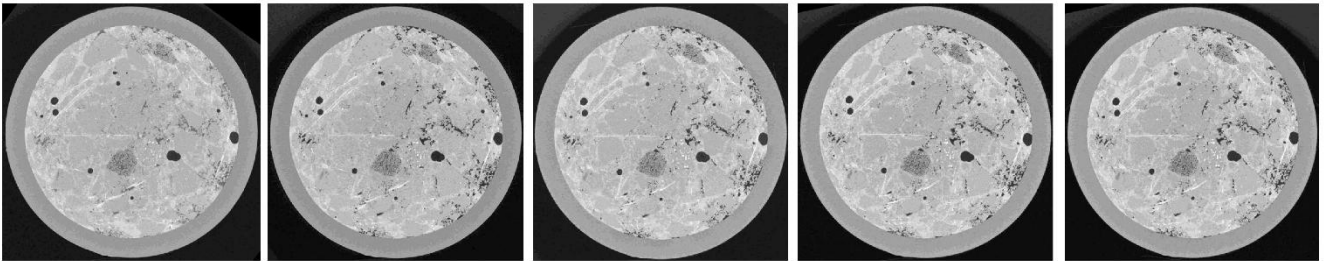
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Figure 2: Slices belonging to plain mortar at different ages. From left to right: 1, 2, 3, 4 and 7-days age.

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Figure 3: Slices belonging to steel fiber-reinforced mortar at different ages. From left to right: 1, 2, 3, 4 and 7-days age.

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The Digital Image Processing (DIP) software AVIZO (FEI Visualization Sciences Group, Hillsboro, Oregon, USA)

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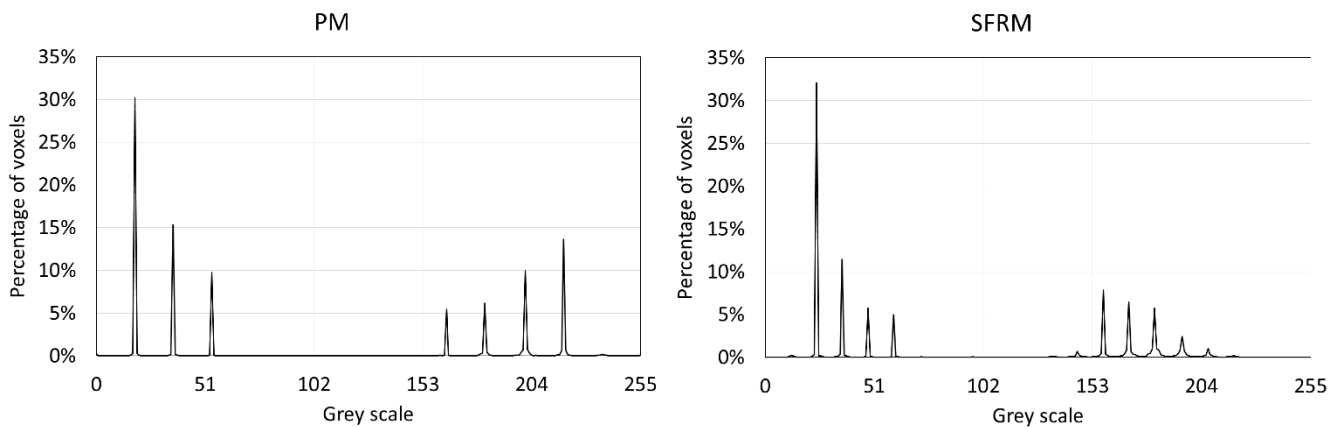
was then used to identify and isolate each individual void inside the specimen. Firstly, the software identified the voxels

142

showing a grey below a certain threshold. In this case, having studied the histogram of grey distribution, the threshold of 65

143

was considered (Figure 4).



144

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Figure 4: Histograms of grey distribution.

146

Figure 4 shows that there is a high percentage of voids below 65, which means empty voxels. However, not all of

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them belongs to voids, most of them are placed in the empty space placed outside the mold. So, the next step was to delete

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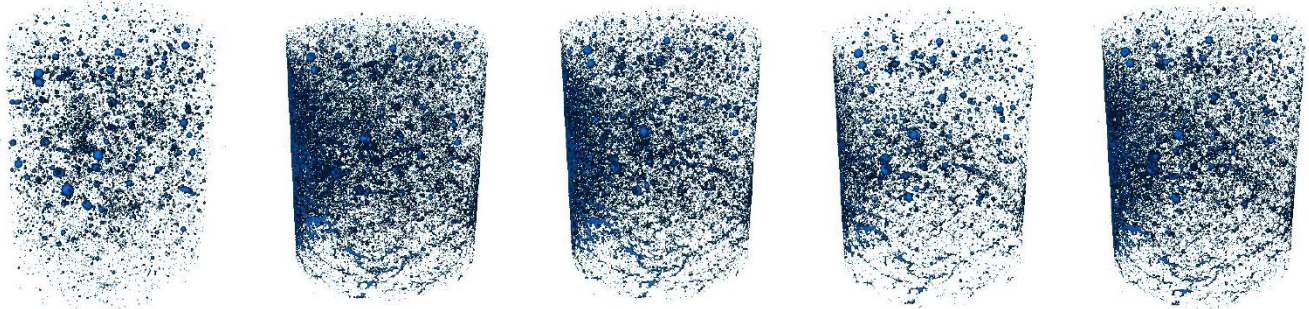
all the voxels which do not belong to the mortar specimen, i.e., the ones placed outside the inner face of the mold. The

149

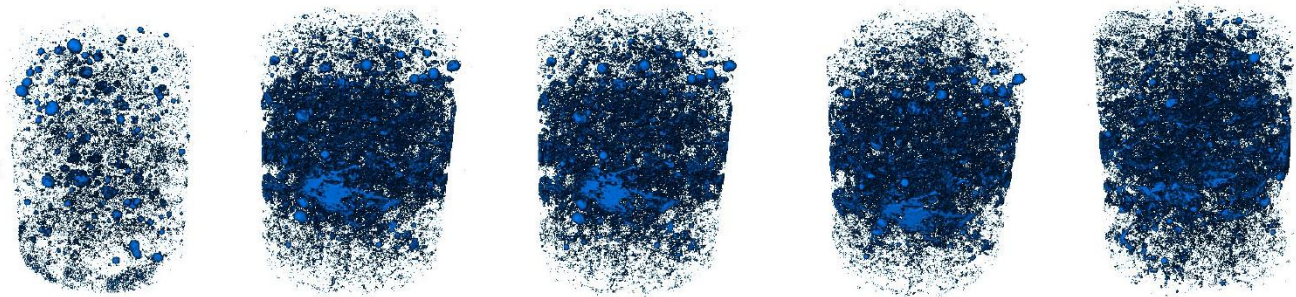
vertical axis of the specimen at each slice is first defined. Next, all the pixels with a distance to the center of the specimen

150 greater than the radius of the specimen are removed. Moreover, the histograms of Figure 4 cannot be used to obtain the
151 percentage of voids in the specimen.

152 Next, all the voxels in contact with each other were grouped, as they represented to the same void. The software
153 identified and isolated the different voids and the final result of the scanning was a dot matrix containing the Cartesian
154 coordinates X, Y and Z of the center of gravity of the pore. Figures 5 and 6 show the image of one specimen of each
155 mixture over time.



156
157 Figure 5: 3D views belonging to plain mortar at different ages. From left to right: 1, 2, 3, 4 and 7-days age.



158
159 Figure 6: 3D views belonging to steel fiber-reinforced mortar at different ages. From left to right: 1, 2, 3, 4 and 7-days age.

160 The first finding of this work is that a relevant number of macropores are placed on the lateral border of the
161 specimen, in contact with the mold, which can be explained by certain phenomena. Firstly, the mortar specimens show a
162 wall effect, leading to a greater percentage of the smallest particles of mortar (water, cement and filler) on the lateral border.
163 Slight mortar shrinkage leads to a gap between mortar and mold, a space that is initially filled with water and then by air.
164 However these “pores” are not the same as voids and may distort the results.

165 Therefore, the lateral region of each specimen was removed, so that instead of the whole specimen, only the inner
166 core of the specimen was analyzed, i.e. a cylinder with a diameter that was 90% of the real diameter of the specimen, i.e.,
167 40.7 mm and the whole height. Additionally, two slices were discarded in both top and bottom faces in order to prevent
168 erroneous results at the very top and bottom ends. In consequence, the studied height was 49.9 mm (Figure 7).

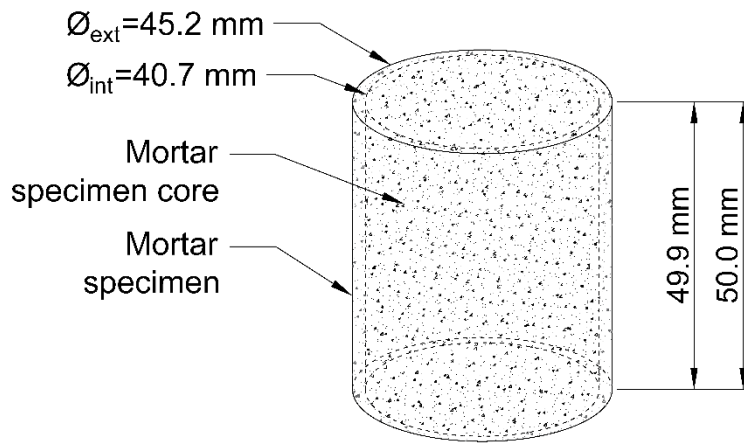


Figure 7: Portion of the mortar to be studied

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170

171 In this work, voids with a less than 3 voxels in the largest direction (i.e., 75 μm length approximately) were discarded
 172 as too small. Moreover, they were not sharply enough defined in the CT-Scan for clear identification. Additionally, pores
 173 larger than 10 mm in the largest direction were discarded as non-representative pores.

174 In consequence, all the micropores [14] are discarded, since they are not possible to be detected using this
 175 technology, and also the smaller portion of the mesopores. In addition, the macropores are also discarded since they
 176 represent an extremely low percentage of the total amount of voids (in all cases less than 8 voids have been detected, which
 177 represents less than 0.01% of the total amount of voids) and a big scatter between the specimens is observed (a difference of
 178 1 void between one specimen and other represents a huge difference in relative terms). In consequence, macropores do not
 179 provide useful information.

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181 3. RESULTS AND DISCUSSION

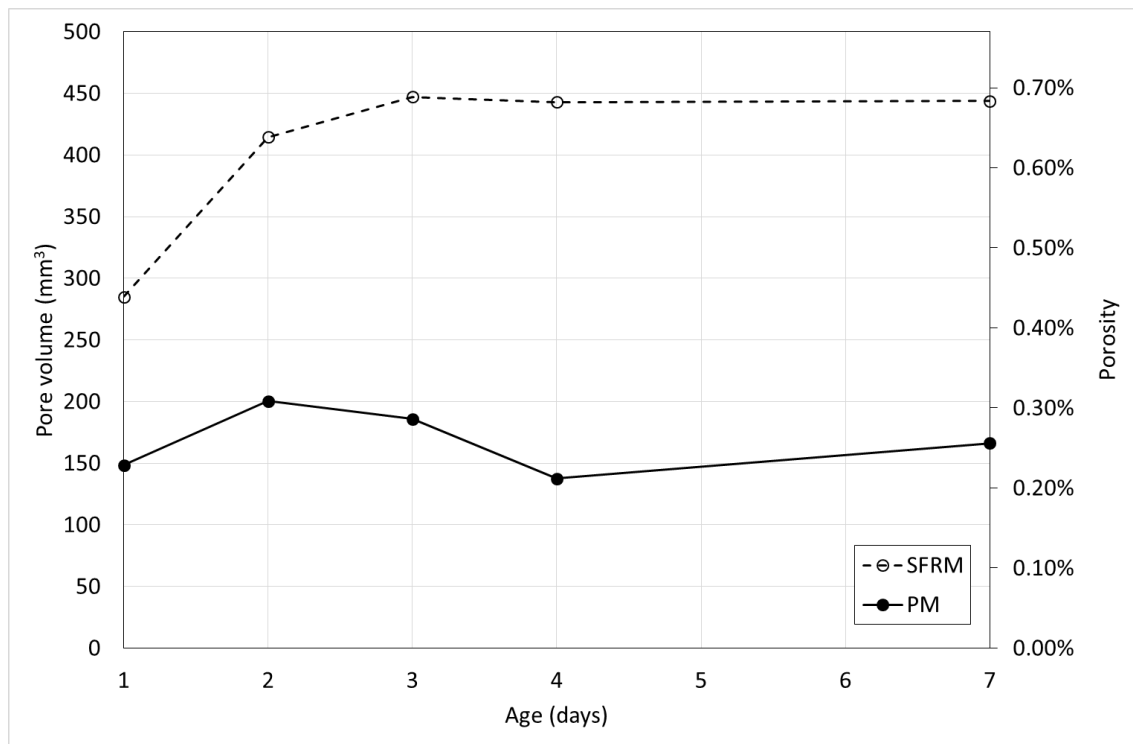
182 A visual inspection of the 3D images is unable to detect relevant differences in the internal porosity between the
 183 specimens nor how they evolve at early ages. The use of DIP software in combination with customized post-processing
 184 routines was required, for a deep analysis of the data and to extract the relevant information. The results from the post-
 185 processing stage are presented in this Section. The values shown in the different tables and figures refer to the average
 186 values of the three specimens belonging to the same mixture. In all cases it has been observed that the coefficient of
 187 variation (COV) is below 10%, which means that the specimens show a low scatter.

188

189 **3.1 Total volume of voids and porosity**

190 The first two parameters to be studied in this research are the total volume of voids and the porosity. This last
191 parameter is defined as the ratio between the total volume of voids and the volume of the specimen considered in this study
192 (i.e. a cylinder with a diameter of 40.7 mm and a height of 49.9 mm, as explained in Section 2).

193 Both parameters were obtained on the 5 different days when the measurements took place over the first week, as
194 mentioned in Section 2 (i.e., 1, 2, 3, 4 and 7 days). Figure 8 shows the changes of the pore volume and the porosity of both
195 mixtures with the age.



196

197 Figure 8: Change of the total volume of voids and the porosity over time of mortar

198

198 The voids inside the specimens can be due to entrapped air and due to the initial water drops that then become voids
199 (since the water is consumed during the hydration process and/or is evaporated).

200

200 Some interesting conclusions can be obtained from Figure 8. The first and most striking conclusion was that the total
201 volume of voids and the porosity was greater in the steel fiber-reinforced mortar (SFRM) than in plain mortar (PM). Since
202 the specimens were not vibrated, a relevant percentage of the voids are because of entrapped air, which tend to be greater
203 than the ones because of initial water drops transformed into voids. These results demonstrate that the presence of fibers
204 results in an increase of the voids content, i.e., the fibers increase the entrapped air.

204

205 They also agree with the results obtained by other researchers [45, 47], who observed that the presence of
206 polypropylene fibers increased the viscosity of the fresh mortar, hindering the removal of entrained air and inducing greater
207 porosity.

208 Regarding the changes in porosity over time, it can be seen that SFRM showed a progressive increase in porosity
209 over time, although it was a damped process. On the contrary, the PM specimens showed no relevant changes over time.
210 The SFRM specimens showed greater porosity than the PM specimens at the beginning of the hardening process (first day).
211 Over the first three days there was a global increase of porosity in both mixtures, although it was more relevant in case of
212 SFRM. Then, the porosity of the PM decreased on the 4th day, following which a slight increase was observed. In case of
213 the SFRM, the porosity remained almost constant after the 3rd day.

214 The dynamic nature of the voids content with two different mechanisms acting in opposing directions was clearly
215 observable. The first one, the hydration process, is a water-consuming activity. At the beginning of the hydration process
216 almost all the voids were full of water and a minimum amount of air was left inside the mortar matrix. As the density of
217 voids full of water is not very different from the one shown by fresh mortar paste, they could not be distinguished in the CT-
218 Scan. Over time and due to the consumption of water during the cement hydration process, the voids empty and dry out,
219 leaving the spaces to be filled by air, which makes them “visible” in the CT-Scan. The curing process therefore results in a
220 progressive increase of the voids content and the porosity of the mortar mixture.

221 On the contrary, the air entrapped inside the voids tends to rise and eventually leave the mortar specimen. In
222 addition, the spaces tend to be occupied by the fresh cement paste, i.e., the voids tend to collapse. This second phenomenon
223 implies a progressive decrease of the void content and the porosity of the mortar mixtures. Macroscopically, the
224 consequence are the autogenous shrinkage and the drying shrinkage.

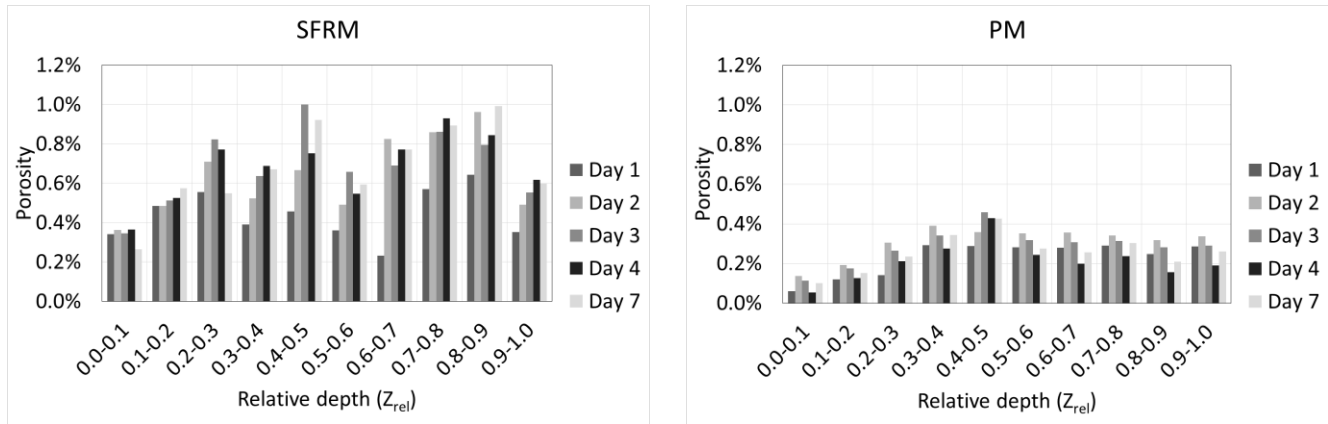
225 The SFRM specimens, because of the presence of fibers, showed a more consistent mortar matrix than the PM
226 specimens. In consequence, the first phenomenon described above prevailed over the second one, resulting in an overall
227 increase of the voids content and porosity over time. On the contrary, the PM showed a fresh and less consistent mortar
228 matrix with both properties equally balanced. In consequence, the void content and the porosity remained almost constant
229 over time [37].

230

231 **3.2 Variation of the total porosity along the depth**

232 Using the coordinates of the center of gravity of each individual void, and more specifically its Z coordinate, it is
233 possible to establish the changes of the voids content and the porosity throughout the depth. Figure 9 shows the change in

234 porosity throughout the depth by the age of the mortar, in both mixtures. In all cases, the depth is shown in relative terms,
 235 i.e., varying from 0 to 1 where 0 refers to the free surface of the specimens and 1 refers to the other end of the specimens,
 236 i.e., the bottom.



237
 238 Figure 9: Change of porosity by depth and by age of SFRM specimen (right) and PM specimen (left)

239 Figure 9 shows some interesting results. First, it may be noted that both SFRM and PM showed a progressive
 240 increase of the porosity throughout their depth. The lowest porosity can be observed in the first two tenths of the specimens.

241 Second, Figure 9 reveals that the SFRM showed, in general, a progressive increase in porosity over time throughout
 242 its depth. This tendency was not observed in the first three tenths and was not evident in the PM specimens where the
 243 porosity remained almost constant over time in all cases.

244 Using the data on porosity over time, the line of the results at each depth was fitted to its slope. This value represents
 245 the average value of the porosity variation speed over the first 7 days. Figure 10 shows the change of this parameter
 246 throughout the depth for both mixtures.

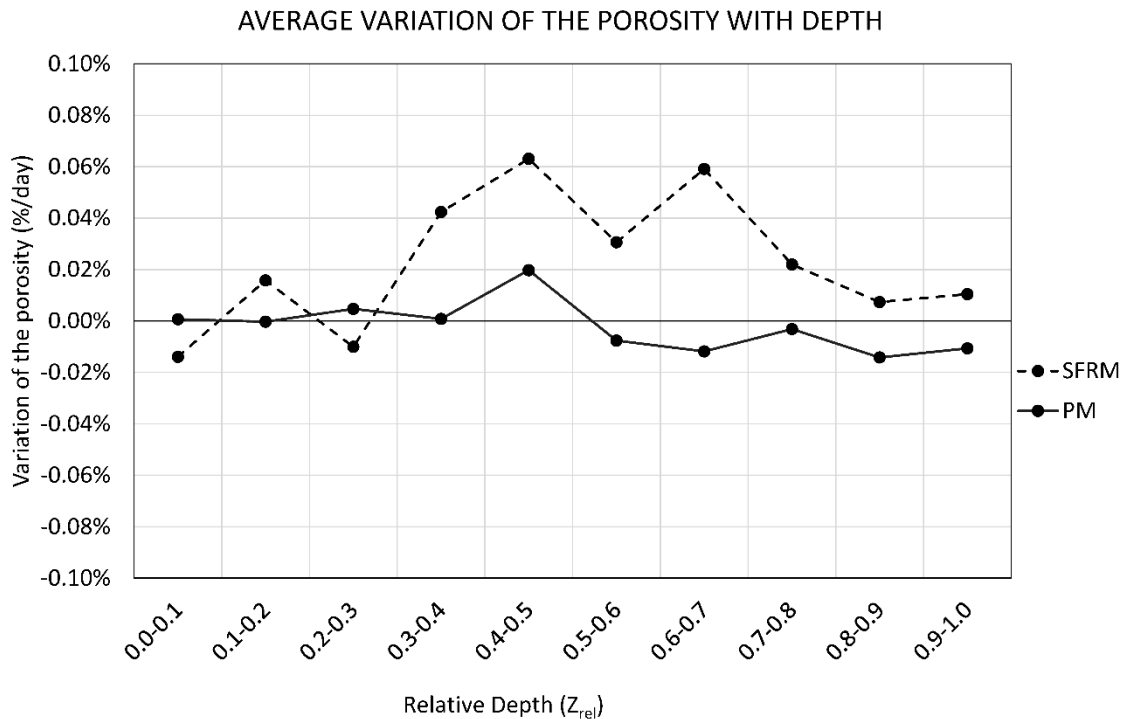


Figure 10: Change of the porosity variation speed along the depth

Figure 10 confirms the findings observed in Figure 9. In general, SFRM showed a positive variation of the porosity over time, while PM showed no relevant changes, i.e., it remained at around zero.

The two opposite mechanisms previously explained can be used to explain these findings. Inside the specimen, i.e., not close to the free surface, the loss of water was mainly due to its consumption during the cement hydration process. In this case, the progressive reduction of free water and, in consequence, the progressive appearance of voids was strongly related to the properties of the cement matrix and its gain in internal stiffness. The water is mostly used to build mortar matrix and, in consequence, to increase the mortar stiffness. Moreover, both phenomena occurred simultaneously, i.e., the loss of water occurred simultaneously with the gains in cement paste stiffness. At the early age, the mortar stiffness is low but most of the voids are full of water (especially the smallest voids), so they are stable. Over time, the voids are emptied but the mortar stiffness increases, so they tend to be stable. In case of the SFRM, the fibers provided extra stiffness to the cement matrix, which prevented the voids from collapsing. On the contrary, the lesser stiffness of the PM cement matrix meant that the voids collapsed with greater frequency.

In contrast, the loss of water close to the free surface was mainly due to evaporation. This phenomenon was unrelated to the properties of the cement matrix and its gain in internal stiffness; moreover, the loss of water happened faster than the gain in cement-paste stiffness. The water is only partially used to build mortar matrix. In consequence, the risk of collapsing voids increased. At the early age, the mortar stiffness is low but most of the voids are full of water (especially the

265 smallest voids), so they are stable. Over time, the voids are emptied and the mortar stiffness does not increase enough, so
266 they tend to be unstable. In this case, the evaporation was relevant and a significant percentage of the specimens were
267 affected by collapse (around 30% of the whole depth), as the specimens were held in an atmospherically controlled room,
268 where they remained at 20°C and 60% humidity,

269 In case of the SFRM, the average porosity variation speed was, approximately, 0.02%/day. On the contrary, in case
270 of the PM, the average porosity variation speed was, approximately, 0.00%/day.

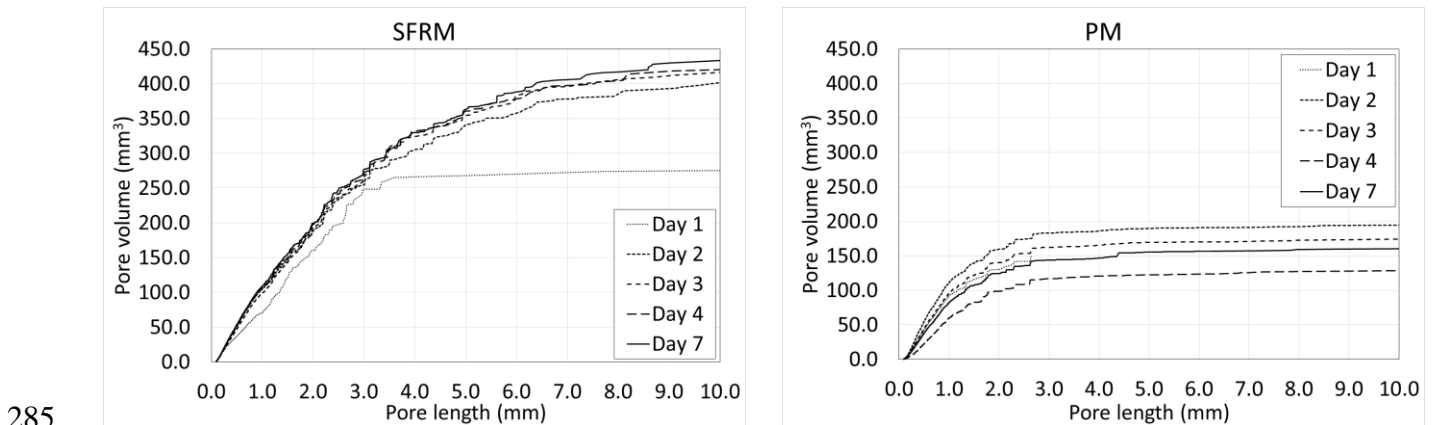
271 In all cases, the specimens showed a slight different behavior around mid-height (i.e., at a relative depth of,
272 approximately, 0.5), because the molds were cast in two parts and showed a higher amount of entrapped air at this height.

273 3.3 Porosity and porosimetric curves

274 The exact geometry of each individual void was obtained through the use of the DIP software. In particular, two
275 geometrical dimensions were of interest: void volume and length. The latter parameter is defined as the maximum distance
276 between two voxels belonging to the same void.

277 Using these two data, the pore volume curves and the porosimetric curves can be plotted. A pore volume curve is
278 defined as the graph that correlates the length of the void and the total pore volume of the voids with a length that is equal or
279 less than the latter. In contrast, the porosimetric curve is defined as the graph showing the correlation between the length of
280 the void and the pore volume of the voids with a length equal to or less than the latter. In both cases, these graphs represent
281 the pore size distribution.

282 Figure 11 shows the pore volume curves of the two mixtures at different ages, while Figure 12 shows the
283 porosimetric curves of both mixtures at different ages. In all cases, the maximum length was limited to 10 mm as the longer
284 pores were residual.



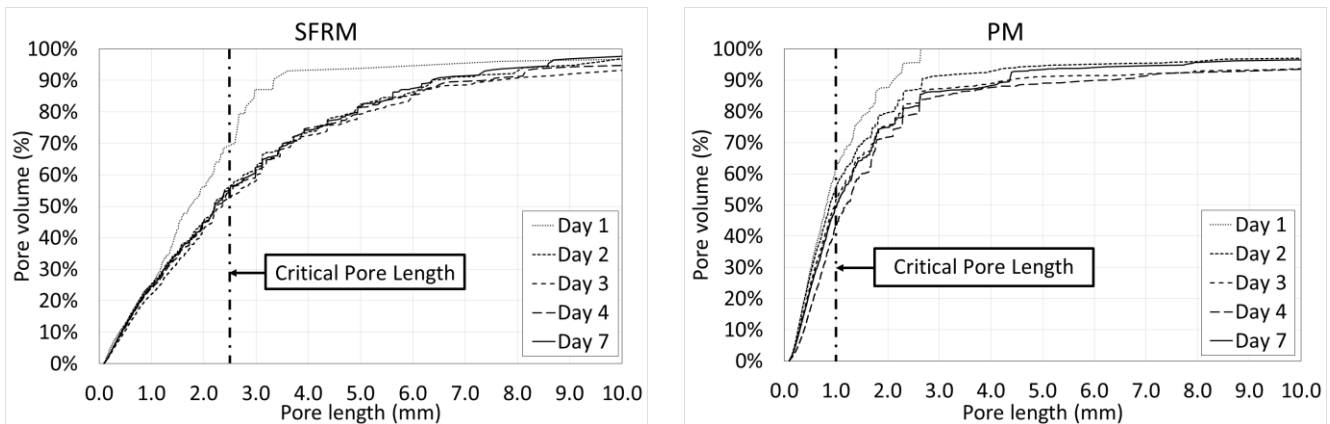
285
286 Figure 11: Pore volume curves of SFRM (right) and PM (left)

287 Figure 11 shows a small change in the pore volume curves by age of mortar and by mixture (SFRM and PM).

288 In case of SFRM, the pore volume curves measured on the first day were notably lower than the other curves. The
289 latter were almost identical. The different behavior of the first curve can easily be explained, since almost all the voids were
290 full of water so were not considered as voids at that time.

291 On the contrary, no clear tendency could be observed in PM specimens, suggesting that neither the total pore volume
292 nor the pore size distribution varied with the age of the mortar.

293



294

295

Figure 12: Porosimetric curves of SFRM (right) and PM (left)

296 In Figure 12, the porosimetric curve of the first day appears, in both cases, above the rest of the curves. This behavior
297 is more clearly observable in SFRM, although it can also be seen in PM specimens. On the remaining days, the porosimetric
298 curves were almost identical, implying no relevant changes in the pore size distribution with the age of the mortar, i.e., the
299 final pore size distribution of the mixture was reached by the second day.

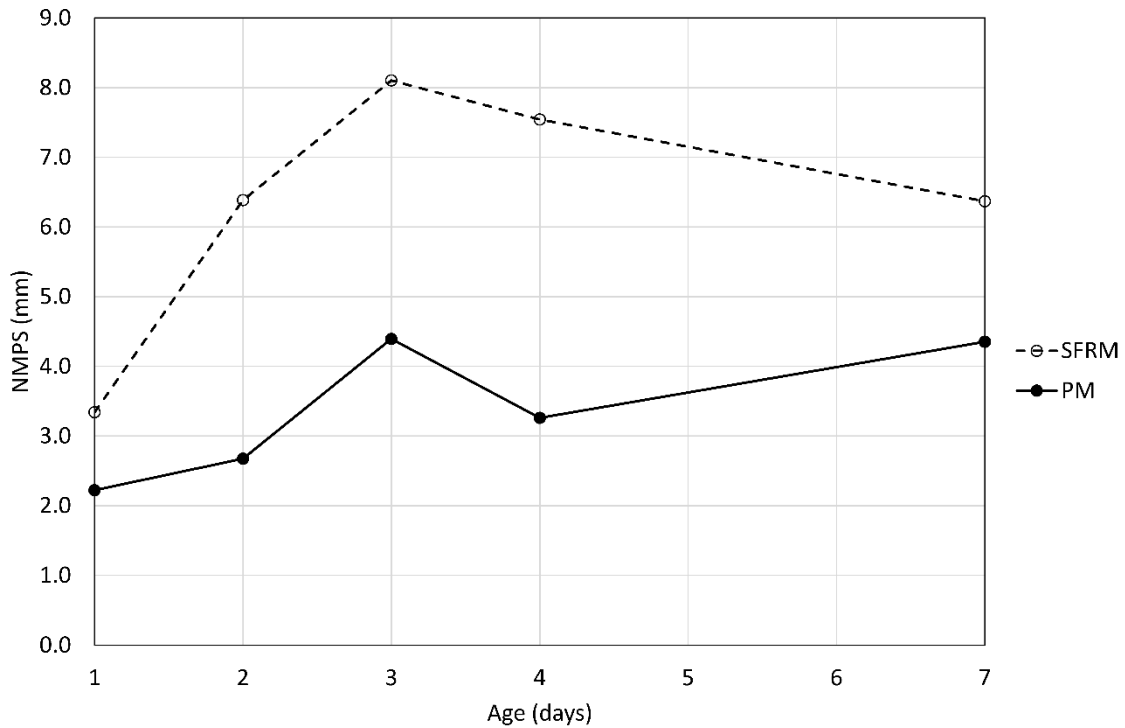
300 Another interesting observation following the comparison in Figure 12 is that porosimetric curves of the SFRM
301 specimens are lower than those of the PM specimens, implying that the SFRM was capable of retaining greater voids than
302 the PM.

303 Both the pore volume curves (Figure 11) and the porosimetric curves (Figure 12) are parabolic with a decreasing
304 slope, meaning that the largest amount of air was retained in the smallest voids, measured in both absolute and relative
305 value. As the pore size (pore length) increased, the capacity of retaining air decreased.

306 Figure 12 also reveals that, in all cases, the curves showed a substantially straight part initially, up to a certain value
307 of pore length, which can be named as “critical pore length”. That is, the critical pore length is defined as the maximum
308 pore length where the slope of the curve pore length – pore volume is constant.

309 In case of SFRM specimen, the critical pore length shows a value between 2 and 3 mm. Beyond this critical pore
310 length, the slope of the curves started to decrease. In case of PM the critical pore length can be established at 1 mm.

311 An interesting parameter that can be obtained through the porosimetric curves is the nominal maximum pore size
312 (NMPS) which can be defined, in a similar way to the well-known nominal maximum aggregate size (NMAS), as the pore
313 length corresponding to a cumulative pore volume of 90%. This value is representative of the pore size distribution, as the
314 remaining 10% of the pore volume corresponds to an extremely low number of individual pores (less than 0.05% of the
315 pores, on average). Figure 13 shows the change of the NMPS in both mixtures over time.



316
317 Figure 13: Change of the NMPS by age.

318 Figure 13 reveals that the NMPS is greater in the SFRM than in the PM specimens at all ages. Hence, the SFRM
319 specimens not only showed a greater pore volume (as can be observed in Figure 8) but also larger voids. In case of the
320 SFRM, there was a relevant increase in the NMPS, up to day 3 after which it started to decrease. However, it can in general
321 be considered that the SFRM specimens showed an increase in NMPS as they aged, which agrees with the results in Figure
322 12. In case of the PM specimens, the curve also shows an increase in the NMPS up until day 3. Subsequently, a decrease
323 occurred on day 4 and a progressive increase. In general terms, it can be considered that the PM showed an increase in the
324 NMPS with the age of mortar, although less intense than in case of the SFRM.

325 All the findings shown above can be explained in terms of a greater stiffness of the SFRM fresh cement paste, due to
326 the additional fiber-induced stiffness, with respect to the PM specimens. In case of the SFRM, its fibers added stiffness to
327 the cement paste and stopped the pores from collapsing. The larger the pore, the greater its instability. In consequence, the
328 SFRM specimens were able to withstand a greater percentage of larger pores than the PM specimens.

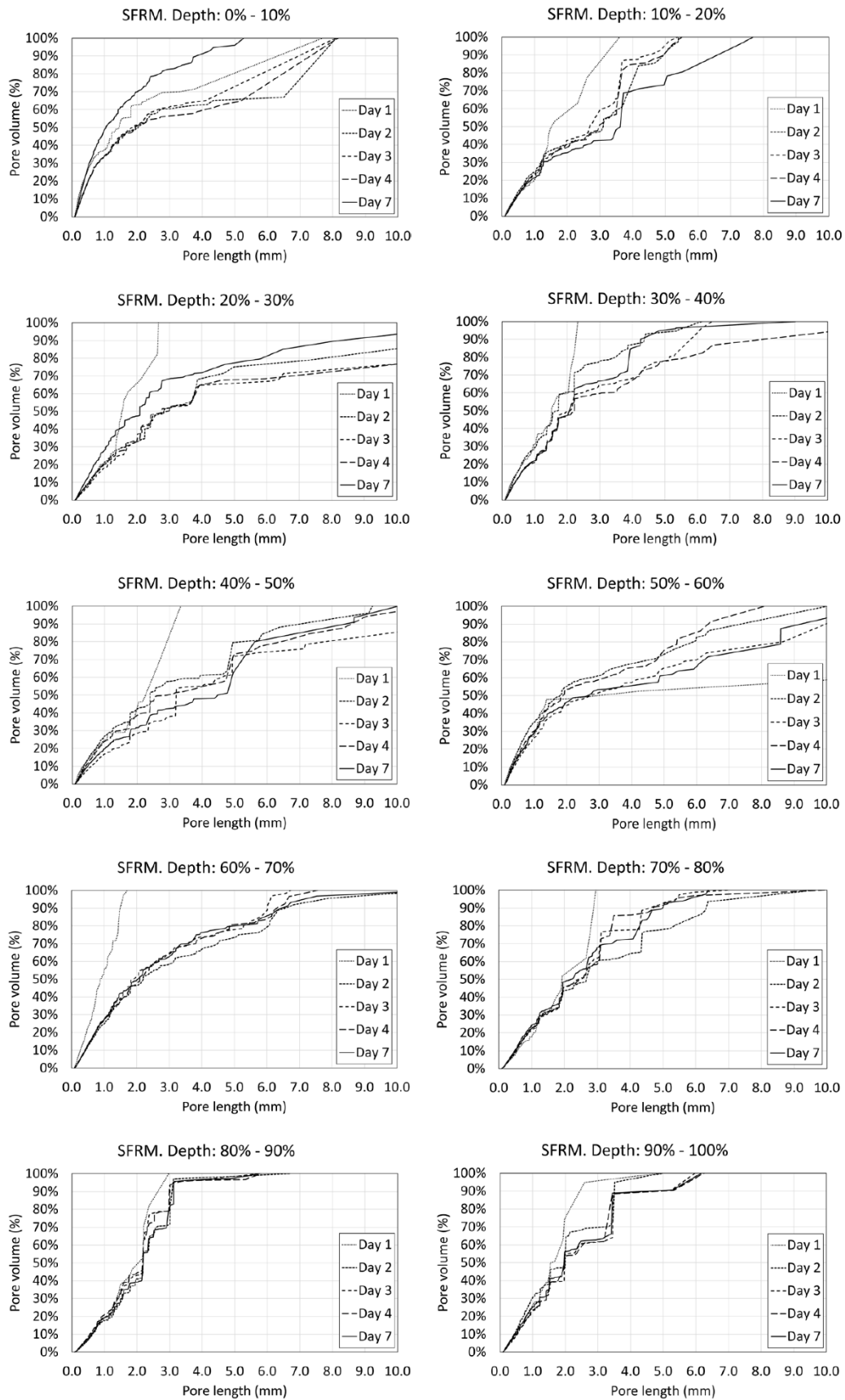
329

330 **3.4 Variation of the porosity and porosimetric curves along the depth**

331 The Z coordinate of the voids is known, from which the porosimetric curves throughout the depth may be estimated.

332 The porosimetric curves are shown in Figures 14 and 15 at different depths. In these figures, the considered depth is top to

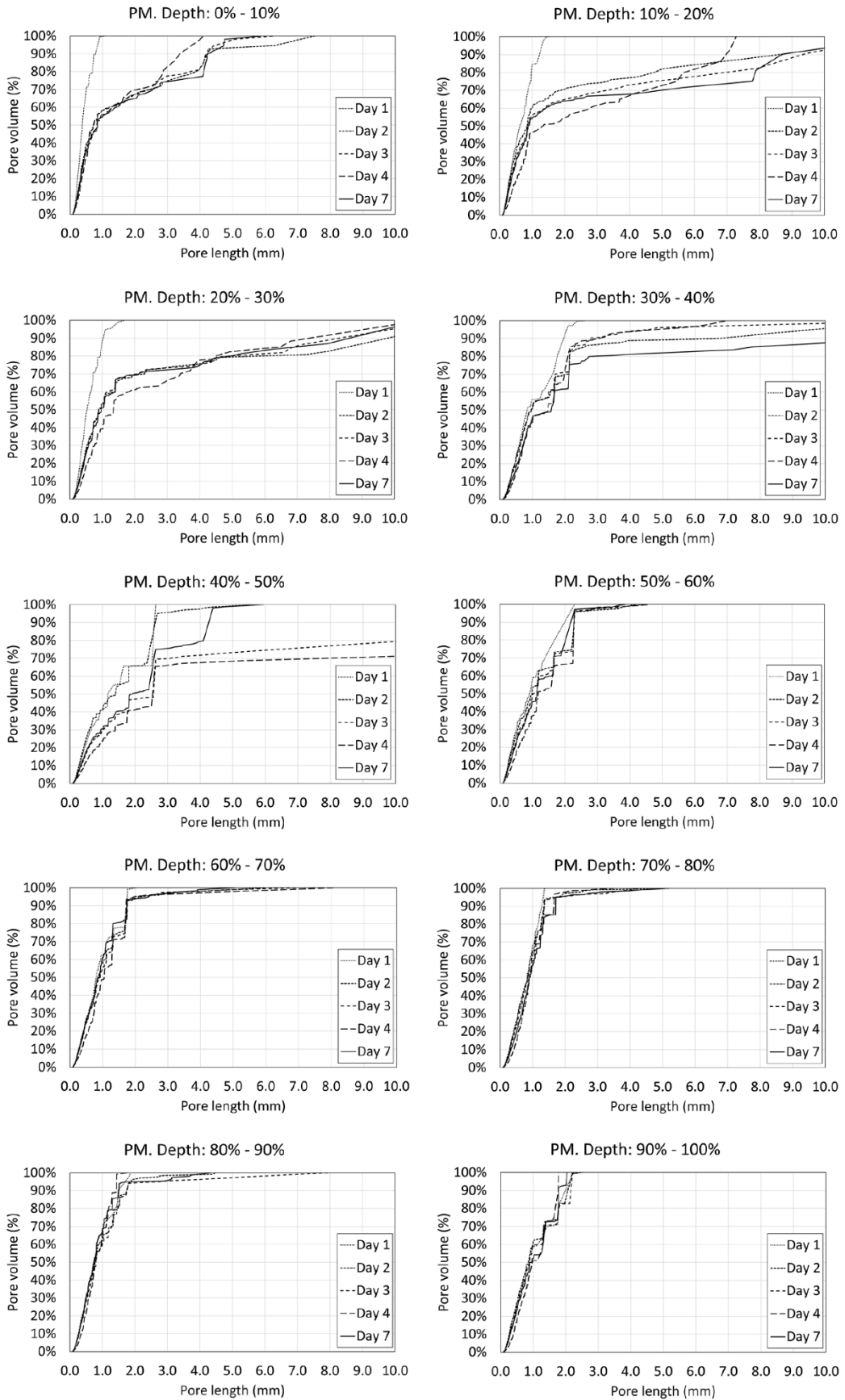
333 bottom.



334

335

Figure 14: Prosimetric curves of SFRM at different age and throughout the depth.



336

337

Figure 15: Porosimetric curves of PM at different age and throughout the depth.

338 Figures 14 and 15 reveal, first of all, that voids are not uniformly distributed throughout the depth. The distance to
339 the free surface, where loss of water occurred, has a strong influence on the pore size distribution.

340 In both mixtures, the curves for the first day can be seen to show a different behavior from the other days. This
341 behavior can be observed at all the depths, because the day 1 voids are, in general, full of water and are neither detected nor
342 identified by CT-Scanning.

343 In case of the SFRM, the upper part of the mixture showed a relevant change of the porosimetric curve with the age
344 of mortar. This variation is explained by the more intense evaporation activity in this area, close to the free surface, which
345 significantly affects the porosimetric curves. In the deepest region of the specimens, far from the free surface, the variation
346 is reduced beyond day 4, when the porosimetric curves started to be almost identical.

347 Something similar was observed at mid-height of the specimens: a relevant change of the porosimetric curves with
348 the age of mortar, due to the molds that were filled in two parts, creating a small horizontal joint in the region, with a
349 relevant amount of entrapped air.

350 In general, all the porosimetric curves showed a first straight part with high slope. This first part went from zero to a
351 certain critical pore length. Beyond the critical pore length, the slope of the curve was significantly lower. The critical pore
352 length varied with the depth and the overall tendency was for the critical pore length to increase with the depth. However,
353 that value was, in general, between 2 and 4 mm. It could also be observed that the critical pore length referred to a
354 cumulative pore volume that increased with the depth, i.e., the first straight part of the curves reached a higher cumulative
355 pore volume with depth: as the depth increased, the percentage of larger pores also decreased. A behavior explained by the
356 hydrostatic pressure of the fresh mortar that increased with the depth. The smallest voids were able to withstand the higher
357 hydrostatic pressure better; beyond the critical pore volume the voids tended to collapse, breaking up into smaller voids.

358 In case of the PM, a relevant temporary variation around the mid-height of the specimens was observed. Except for
359 the first day (where many voids are filled with free water and undetectable in the CT-Scan), the tendency is towards a
360 progressive reduction of the greater pores.

361 In a similar way to the SFRM specimens, all the PM curves were substantially bi-linear. The first part was
362 approximately a straight line with a large slope up to a critical pore length. Beyond this value, the slope of the curve
363 decreased drastically. In all cases, the lines of the first part of the curve showed a similar slope. However, the critical pore
364 length varied between 1 and 3 mm with the depth. Hence, the percentage of larger pores decreased with the depth. The
365 explanation is the same as in case of the SFRM: an inverse relation exists between the hydrostatic pressure of fresh mortar
366 and the maximum stable void volume. The voids smaller to this critical volume were stable, but the larger voids tended to
367 collapse, breaking up into smaller voids.

368 The slope of the curve, directly related with the viscosity and stiffness of the cement paste [45, 47], was significantly
 369 steeper than in SFRM. The lesser stiffness of the PM cement paste caused larger voids to become more unstable than in
 370 SFRM specimens and tended to collapse, breaking up into smaller voids.

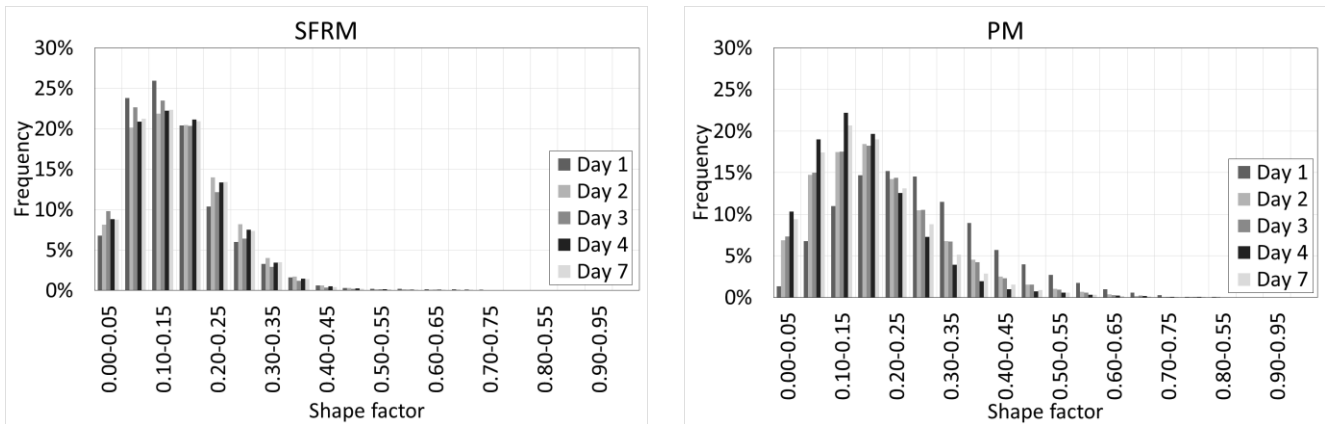
371 3.5 Shape factor of the voids.

372 As explained before, using the data provided by the CT-Scan and the DIP software, it is possible to establish the
 373 volume and the length of each void. Then, the shape factor of each pore can be obtained, defined as the quotient between the
 374 volume of the pore and the volume of the sphere circumscribed to the pore (eq. 1).

$$SF = \frac{V_p}{\frac{1}{6} \cdot \pi \cdot L_p^3} \quad (1)$$

375 where V_p is the pore volume and L_p is the pore length.

376 Figure 16 shows the histograms of the shape factor of the different mixtures.



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 378 Figure 16: Histogram of the shape factor of SFRM (right) and PM (left)

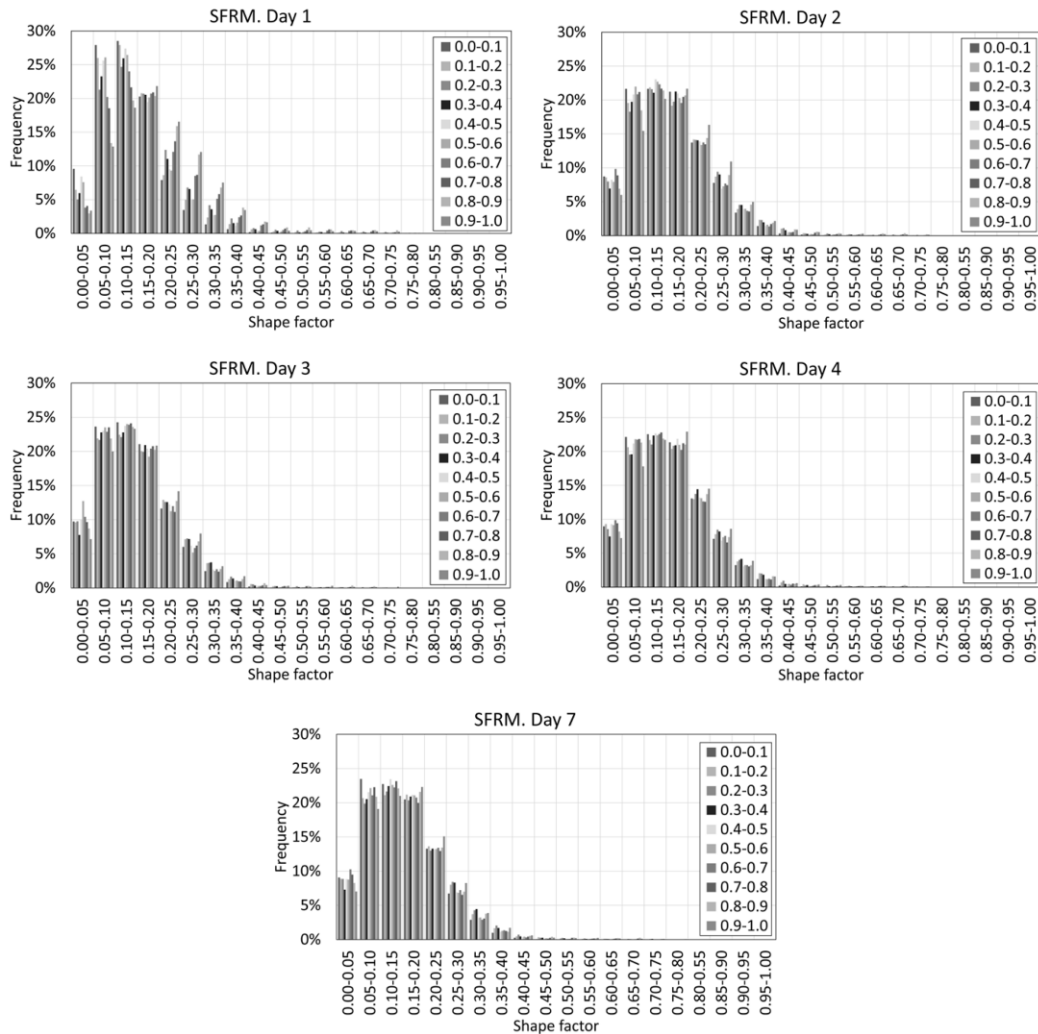
379 Some interesting observations can be drawn from Figure 16. First of all, it can be seen that the voids were far from
 380 spherical in all cases, as the shape factor was far away from 1. The SFRM specimens showed even smaller shape factors
 381 than the PM specimens, with a mode value between 0.10 and 0.15 and over 90% of the voids showed a shape factor below
 382 0.30.

383 The PM specimens showed higher shape factor values than the SFRM, with a mode value between 0.15 and 0.20 and
 384 more than 90% of the voids with a shape factor below 0.40.

385 These values are explained by the fact that the voids are formed in cement paste components of lesser stiffness and
 386 tend to occupy the spaces that are not occupied by the other cement paste components. These spaces tend not to be
 387 spherical, but flaky and elongated. In case of the SFRM, the spaces tend to be even flakier and/or more elongated because of
 388 the fiber presence. Once again, the fibers modified the pore morphology.

389 Regarding the variations of the shape factor histogram, it may be seen that, the histograms of the SFRM specimens
 390 move toward a larger shape factor as the age of the mortar increases. On the contrary, the PM histograms move toward
 391 smaller shape factors as the age of the mortar increases. However, in both cases this variation is not very relevant.

392 The shape factor showed relevant changes throughout the depth. The shape factor histograms for both mixtures are
 393 shown in Figures 17 and 18 at different ages and at different depths.



394
 395 Figure 17: SFRM shape factor histograms by depth and by the age of mortar.

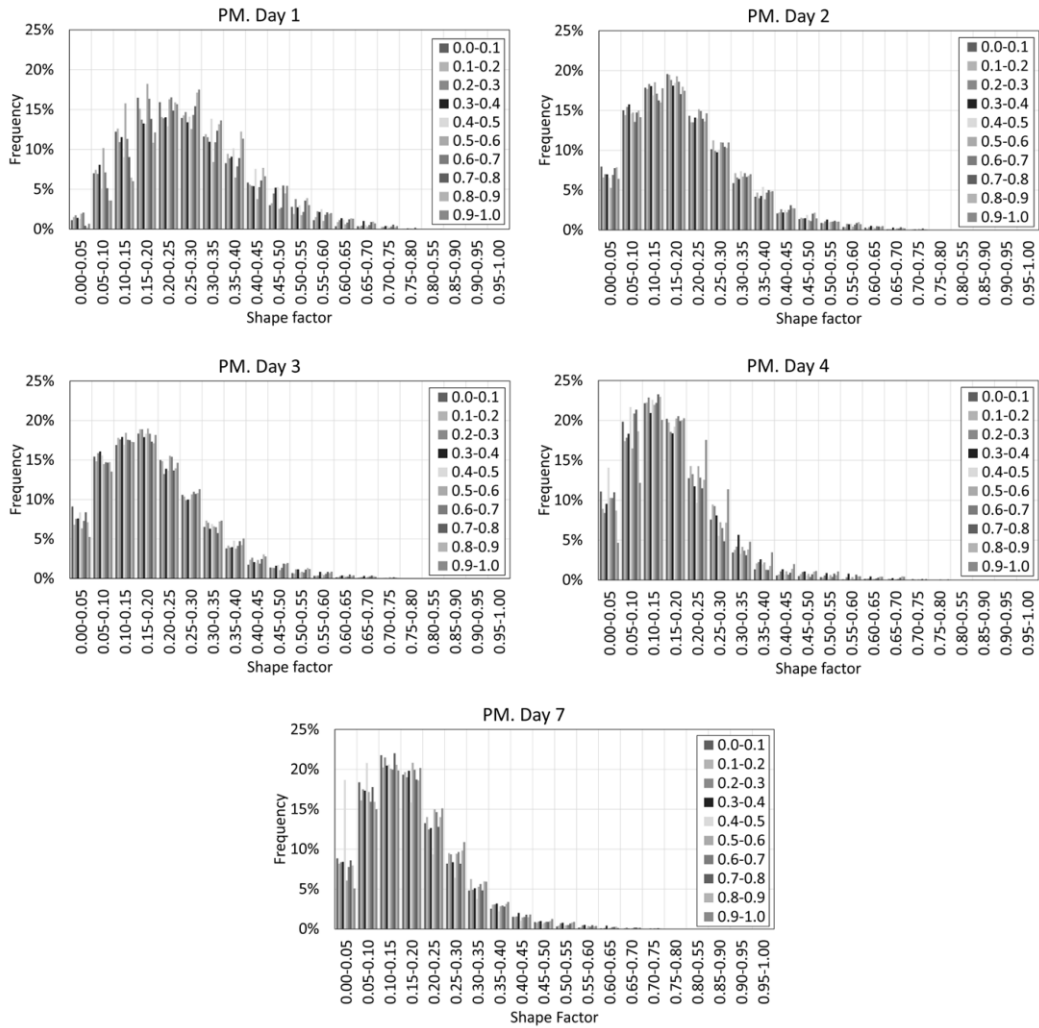


Figure 18: PM shape factor histograms by depth and by the age of mortar.

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In Figures 17 and 18, day 1 shows a slightly different behavior in both cases. In case of the SFRM, the shape factor histogram for day 1 is positioned slightly to the left, i.e., a smaller shape factor. This smaller size implies that the voids show small shape factors at very early ages and then increase with the age of the mortar. In case of the PM, the difference between the shape factor histogram for day 1 and the other days was more notable and in an opposite direction, i.e., the shape factor histogram for day 1 is positioned slightly to the right, i.e., a higher shape factor. Hence, at very early ages the voids show higher shape factors and then decrease with the age of the mortar.

Regarding the variation of the shape factor by depth it can be seen that this change was very small in SFRM specimens. Day 1 deserves a special mention, where it is clearly observed that the shape factor histogram moves toward the right as the depth increases, i.e., the shape factor increases with the depth; a variation that is hardly observable over the other days.

408 In case of the PM specimens, a small movement of the shape factor histograms toward the right was observed, i.e., a
409 slight reduction of the percentage of voids with the smallest shape factor and, in consequence, a slight increase of the
410 percentage of voids with higher shape factors. This increase was due to the hydrostatic pressure of fresh cement paste,
411 which promoted the creation of voids with a more stable geometry, i.e., more spherical.

412 However, in PM specimens, the change of the shape factor histograms tended to be a slightly chaotic with the age of
413 the mortar. The collapse of the pores, as explained previously, was due to the lesser stiffness of the fresh cement past, which
414 changed the shapes of the voids over time. These changes were clearly observed on day 4, where there was a relevant
415 increase in the percentage of pores with smaller shape factors.

416

417 **4. CONCLUSIONS**

418 The change of the internal pore morphology of two different mixtures has been analyzed throughout the first week of
419 curing, using a CT-Scan and DIP software. Both mixtures showed the same components and dosage of the cement matrix
420 and the only difference between them was that the SFRM mixture incorporated 0.1% by volume of steel fibers, while the
421 PM mixture included no fibers.

422 The pore morphology was measured during the first week of curing, when most of the variations within the mortar
423 matrix are expected to occur, at 5 different ages, i.e., 1, 2, 3, 4 and 7 days.

424 Some interesting conclusions have been obtained. First, it can be noted that the SFRM specimens showed greater
425 porosity than the PM specimens. This difference was observed over all the days under study. The porosity increased with
426 the age of mortar in both cases, although it was more relevant in SFRM specimens.

427 Most of these observations can be explained by the stiffness of the fresh cement paste. At the beginning of the
428 hardening process almost all the voids are full of water. Over time, the water is consumed and the space occupied initially
429 by water is then filled with air, and the voids appear. There are two water-consumption mechanisms and, in consequence,
430 two ways of creation of voids. The first mechanism is the hydration process, which has a high water-consumption rate
431 affecting the whole specimen. The second mechanism is water evaporation, which only occurs close to the free surface.

432 With regard to the first phenomenon, there was a strong relation between the creation of voids and the increased
433 stiffness of the fresh cement paste. In case of the SFRM specimens, the extra fiber-induced stiffness permitted the retention
434 of larger voids. In case of the PM specimens, the absence of this extra stiffness meant that the larger voids tended to
435 collapse.

436 Water loss happened more quickly closer to the free surface and not all of it was used to create the cement paste
437 (because of water evaporation). The consequence was a lower mortar stiffness close to free surface that could only retain
438 smaller voids. Even in case of the SFRM specimens, the fibers provided no extra stiffness in an efficient way.

439 The porosimetric curves showed two different parts. The first one belonging to the smaller sizes, went from 0 to a
440 critical length. In this part, the curves tended towards a straight line and beyond that critical length, the slope of the curves
441 decreased. The critical length of the SFRM specimens was defined between 2 and 3 mm, while the critical length was
442 around 1 mm for the PM specimens. This behavior was observed throughout the depth, although the value of the critical
443 length varied slightly with the depth, especially in SFRM specimens.

444 The critical length is, again, closely related to the stiffness of the fresh cement paste and its capacity to develop
445 voids. In case of the SFRM specimens, the extra fiber-induced stiffness permits higher retention of mortar pores of up to 2-
446 to-3 mm in length. Beyond this value, the pores became more instable and tended to collapse, breaking down into smaller
447 ones. In case of the PM specimens, the lower stiffness of the cement paste and its capacity to retain pores was reduced;
448 voids beyond 1 mm length were only retained with difficulty and tended to collapse.

449 Some interesting conclusions have also been drawn from the shape factor histograms. First of all, it can be noted that
450 rather than spherical, the pores were flaky and elongated. The shape factor of the SFRM specimens due to the presence of
451 fibers was smaller than the PM shape factor. In case of the SFRM, the shape factor increased with both the age and the
452 depth of the mortar. On the contrary, in case of the PM, the shape factor decreased with the age of the mortar, although it
453 increased with the depth. However, the change was in all cases less relevant.

454 It has therefore been demonstrated in this study that fibers modify several aspects of pore morphology: they increase
455 mortar porosity, increase mortar pore size and reduce the shape factor, among other effects. This should be used in the
456 design of mortar mixtures, especially when a certain of entrained/entrapped air is required.

457

458 **5. ACKNOWLEDGEMENTS**

459 The authors are grateful for the financial support from the Ministerio de Economía y Competitividad, BIA2015-
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461

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